# Chemical Structure and Properties of Polyimides Derived from Asymmetric Diamines（p－ODA and p－TPEQ） 

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## 1．Introduction

Aromatic polyimides（PIs）are attractive for aerospace applications because of their excellent thermal and envirommental stabilities，and high resistant properties．Previously．Yokota et al． reported that an asymmetric PI prepared from asymmetric and nonplanar biphenyl tetracarboxylic dianhydride with 4．4＇diaminodiphenyl ether （4．4＇－ODA）have a higher glass trausition temperature（ $\mathrm{T}_{\mathrm{g}}$ ）and molecular mobility above $\mathrm{T}_{\mathrm{g}}$ than those of symmetric PI（ s －BPDA－PI）．We also reported that a PI film derived from an asymmetric of oxydiphthalic dianhydride，namely a－ODPA with a flexible ether linkage exhibited higher $\mathrm{T}_{\mathrm{g}}$ and high molecular mobility at temperature region above $\mathrm{T}_{\mathrm{g}}$ than than those of s－ODPA－PI．Moreover，PI（a－ODPA／4，4＇－ODA）was found to have excellent heat－sealing properties，i．e．it could be found to be heat－sealed within a few minutes at $350{ }^{\circ} \mathrm{C}$ with maintaining the original strength．The increased $\mathrm{T}_{\mathrm{g}}$ ，high molecular mobility and thermal adhesiveness can be explained from the suppressed internal rotation of ether linkage between two phthalimide groups． However，the detailed thermal and mechanical properties of PI films derived from asymmetric diamines have not been studied enough．
In this study，we report the effects of chemical structure of PIs derived from novel asymmetric diamines：2－phenyl－4－4＇－diaminodiphenyl ether（p－ODA）and 2－phenyl－1，4－bis（4－aminophenoxy）benzene （ p －TPEQ）which have a phenyl group as a side chain on the thermal and mechanical properties of their films

## 2．Experimental

After the preparation of the DMAc solution in which corresponding diamines were dissolved dianhydride was added at room temperature．The heterogeneous solution was stirred for 6 hours at room temperature to afford the polyamic acids（PAA）．The molecular weight of PAA were adjacent to almost $0.5(\mathrm{dL} / \mathrm{g})$ with an addition of phthalic anhydride（PA）at a $15 \mathrm{wt} \%$ solids concentration．PAAs were
converted to the corresponding PIs by typical method of thermal imidization. PAA were cast onto gla plate, dried in oven at $60^{\circ} \mathrm{C}$ for 2 hours, and then cured up to $250^{\circ} \mathrm{C}$ by stepwise heating of $50^{\circ} \mathrm{C} / 0.5$ $150^{\circ} \mathrm{C} 0.5 \mathrm{~h}+200^{\circ} \mathrm{C} / 0.5 \mathrm{~h}+240^{\circ} \mathrm{C} / 0.5 \mathrm{~h}$ onto glass plate under vacuum. Then these PAA films we removed from the substrate. These PAA films were wrapped in aluminum and cured up to $350^{\circ} \mathrm{C} 1$ stepwise heating of $250^{\circ} \mathrm{C} / 0.5 \mathrm{~h}\left(+300^{\circ} \mathrm{C} / 0.5 \mathrm{~h}+350^{\circ} \mathrm{C} / 0.5 \mathrm{~h}\right)$ under vacuum.

## 3. Results and discussion

Fig. 2 and Figu. 3 show the DMA data of various PI films to compare the effect of asymmett structures of diamine on the molecular motilities of PI chains. PI( $\mathrm{s}-\mathrm{BPDA} / 4,4^{*}$-ODA) at PI(s-BPDA/1,4,4-APB) films show the small drops of $E$ above $T_{g}$. indicating that these PIs formed his


Fig.2. DMA curves of p-ODA-PIs


Fig.4. DMA curves of s-BPDA/p-TPEQ


Fig.6. DMA curves of s -BPDA/p-ODA


Fig.3. DMA curves of p-TPEQ-PIs


Fig.5. DMA curves of a-BPDA/p-TPEQ


Fig.7. DMA curves of PMDA/p-ODA
ordered structures attributed to the symmetric and planar structures of s-BPDA. On the other hand. PIs derived from p-ODA or p-TPEQ showed the large drops of $E$ ' above $T_{g}$, indicating that these PIs form amorphous structures. On the basis of above results, the formation of amorphous PIs was found to be mainly depended on the asymmetricity of dianhydride.

Fig. 4-6 show the DMA data of PI(s-BPDA/p-TPEQ), PI(a-BPDA/p-TPEQ) and PI(s-BPDA/p-ODA) cured at the different temperatures. In general, aromatic PIs with symmetric and planar structures were proceeded to the formation of ordered structure by curing at higher temperature. On the other hand, asymmetric PIs were not thought to be proceeded to form ordered structure by curing due to their amorphous structure. However, PIs derived from asymmetric diamine observed in Figure 4-6 seem to have a trend of extending the rubbery plateau region by curing at higher temperature. Similar phenomena were also observed in the case of PIs derived from a-BPDA with symmetric diamine ( $4,4^{*}-\mathrm{ODA}$ or $1,4,4-\mathrm{APB}$ ). These behaviors are thought to be depended on a tangle of polymer chains and polymer structure in the amorphous regions. These detail observations are now under considerations in terms of a difference in molecular weight and by using of wide-angle X -ray diffraction measurements.

In addition, Fig. 7 shows the DMA data of $\mathrm{PI}(\mathrm{PMDA} / \mathrm{p}-\mathrm{ODA})$. $\mathrm{PI}(\mathrm{PMDA} / \mathrm{p}-\mathrm{ODA})$ film also showed the large drop of $E^{\prime}$ to $10^{7} \mathrm{~Pa}$ above $\mathrm{T}_{\mathrm{g}}$. indicating that asymmetric and nonplanar structure of p -ODA, and steric hindrance of phenyl group were found to cause PMDA-based PIs to give a high molecular mobility.

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polyimides had good solubility both in polar aprotic sovents and in some commom organic solvents. All the polyimides were amorphous with high $\mathrm{T}_{\mathrm{g}}$ and exhibited excellent thermal stability and mechanical properties.

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